A low temperature cluster condensation approach to CdS nanocrystals: oxidative aggregation of $[Cd_{10}S_4Br_4(SR)_{12}]^{4-}$ with sulfur[†]

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The cluster salt [N(C₂H₅)₄]₄[Cd₁₀S₄Br₄(Sp-Tol)₁₂] reacts with six equivalents of sulfur in dimethylformamide at 140 °C to produce polycrystalline (wurtzite type) CdS nanoparticles of 5.8 nm mean diameter, which on their surface are ligated with p-tolylthiolate ligands, dimethylformamide and water; UV/vis, PL, ¹H- and ¹¹³Cd-NMR spectra recorded during various stages of the synthesis indicate that the CdS nanoparticles are formed by a cluster condensation process.

The theoretical concept of using molecular clusters as synthetic precursors for the construction of crystalline inorganic solids is rooted in the recognition that certain transition element clusters contain structural fragments of the respective crystalline bulk phases.¹⁻³ The condensation of suitable clusters could offer a low energy pathway to crystalline solids, which otherwise are only accessible by high temperature solid state reactions. Despite the logic of this approach, examples for the successful application of this synthetic concept are extremely rare. Two examples are the reported condensation of [Mo₃S₁₃]²⁻ clusters to give crystalline MoS₂ nanofibers⁴ and the thermolysis of $[M_{10}Se_4(SPh)_{16}]^{4-}$ clusters (M = Cd, Zn) to produce CdSe and ZnSe nanocrystals.⁵ In order to test the usefulness of the readily accessible [Cd₁₀S₄Br₄(Sp-Tol)₁₂]⁴⁻ cluster anion⁶ as a precursor to wurtzite phase CdS nanocrystals, we investigated its reactions with elemental sulfur in dimethylformamide/carbon disulfide. Because the tetra-adamantane core of the Cd₁₀ cluster anion is a fragment of the cubic CdS (sphalerite) structure, oxidatively induced aggregation of this unit was expected to yield crystalline CdS nanoparticles under mild conditions.‡

When a solution of six equivalents of elemental sulfur in 0.5 mL carbon disulfide is added to a heated solution of 50 mg of the cluster salt (Et₄N)₄[Cd₁₀S₄Br₄(Sp-Tol)₁₂] (1, 140 °C) in 5 mL of DMF an immediate color change of the solution from pale to intense yellow indicates rapid aggregation of the cluster anion and formation of CdS nanocrystals. After 10 minutes, the reaction mixture is cooled to room temperature, and the nanocrystals are precipitated as a yellow-orange solid by addition of tetrahydrofuran. The nanocrystal yield is 87% based on Cd.

A transmission electron micrograph of the crude CdS nanocrystals is shown in Fig. 1 together with their size distribution, as determined by optical analysis of the micrographs. The particles have a mean diameter of 5.8 nm (std. 20%), and, due to the absence of organic surfactants, form weakly bonded aggregates that are readily soluble in polar solvents such as pyridine and DMF. The X-ray powder spectrum of the nanocrystals (ESI†) shows broad peaks typical of the hexagonal (wurtzite) form of CdS (JPS card 41-1049). Based on the half-width of the 110 reflection, the mean crystallite size of the particles is calculated as 3.5 nm using the Debye–Scherrer formula. This value is smaller than the TEM diameter of 5.8 nm, but still within the error margins of both experiments. The UV/vis spectrum of the nanocrystals in DMF reveals an absorption edge of 430 nm, which corresponds to the

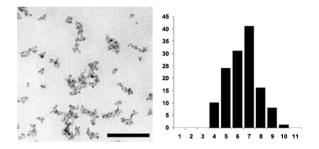


Fig. 1 Transmission electron micrograph of CdS nanocrystals formed by reaction of $\bf 1$ with sulfur in hot DMF. Scale bar is 100 nm. A particle size histogram is also shown.

band gap excitation (2.88 eV). A comparison with the value for bulk CdS (2.42 eV)⁸ shows that the CdS nanocrystals are in the size-confinement regime. The photoemission spectrum of the nanocrystals (inset of Fig. 2, curve C) looks remarkably similar to that of single nanocrystals⁹ and has a maximum at unusually low energy, 492 nm. CdS nanocrystals of comparable size usually emit in the 440–460 nm range. ¹⁰ Both effects are likely consequences of the presence of residual Cd₁₀S₄ fragments on the nanocrystal surface which provide trapping sites below the conductance band.

On the basis of their IR spectra (ESI), the as-synthesized CdS nanocrystals are ligated by a mixture of DMF, water, and *p*-tolylthiolate molecules. Bands around 2900–3000 cm⁻¹ are dominated by the C–H stretching vibrations of the DMF. The unsymmetrical O–H vibration of bound water (3600–3100 cm⁻¹) partially overlaps with this band. Evidence for the presence of coordinated thiocresol comes from three bands at 1082, 1013, 802 cm⁻¹, which correspond to the ring vibration of 1,4-disubstituted benzene, and which can be observed at virtually the same positions in the free thiol. The C=O vibration

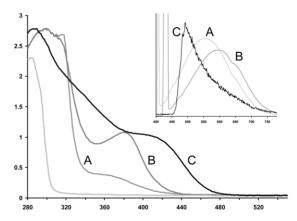


Fig. 2 UV/vis spectra (300 MHz, DMF-d₇) of A) **1** before heating, B) **1** after 1 h at 100 °C, C) purified CdS nanocrystals in DMF. The spectrum of 4-methylphenylthiol (unmarked) is shown for comparison. Inset: emission spectra of **1** A) before heating ($\lambda_{\rm ex} = 400$ nm), B) after 1 h at 100 °C ($\lambda_{\rm ex} = 429$ nm), and C) purified CdS nanocrystals in DMF ($\lambda_{\rm ex} = 300$ nm).

 $[\]dagger$ Electronic supplementary information (ESI) available: X-ray powder, IR and ^{13}C NMR spectra. See http://www.rsc.org/suppdata/cc/b3/b302266h/

of DMF appears as a strong band shifted to $1636~{\rm cm}^{-1}$ as is typical for coordinated DMF.

In order to elucidate the mechanism of nanocrystal formation, UV/vis. PL. ¹H NMR and ¹¹³Cd NMR spectra were recorded during various stages of the nanocrystal synthesis. In DMF solution, the [Cd₁₀S₄Br₄(SR)₁₂]⁴⁻ cluster anion gives rise to a broad absorption shoulder at 364 nm (Fig. 2, spectrum A) and a luminescence with a maximum at 557 nm (inset, spectrum A), in agreement with earlier measurements.6 The ¹H NMR spectrum (Fig. 3A) of the cluster salt contains doublets of o- and *m*-protons of cluster bound *p*-tolylthiolate ligands. However, additional weak pairs of doublets indicate that the cluster anion undergoes substitution of bromide by DMF to produce the species $[Cd_{10}S_4Br_{4-x}(DMF)_x(SR)_{12}]^{(4-x)-}$ (x = 1...4). Heating of the cluster solution to 100 °C under oxygen-free conditions for 5 min and then for 1 h (spectra B and C) completes the ligand substitution to produce the neutral cluster $[Cd_{10}S_4(DMF)_4(SR)_{12}]$ (doublets at 7.28 and 7.63 ppm for mand o-protons of p-toluylthiolate ligands) traces of which can already be seen in spectrum A. Cases of intra- and intermolecular exchange of the terminal ligands have been documented for related Cd complexes and clusters. 11,12 The emission characteristics of the DMF ligated cluster ($\lambda_{max} = 600$ nm, B in insert of Fig. 2) are as expected very similar to 1, whereas the absorption at 364 nm (Fig. 2B) has grown in intensity and slightly shifted to 382 nm.

The acquisition of ¹¹³Cd NMR data on **1** in DMF was problematic because of the low solubility of the salt in this solvent. In DMSO solution, however, **1** exhibits signals at 567 and 651 ppm [*vs.* aqueous Cd(ClO₄)₂], for the four outer and six inner Cd²⁺ sites respectively (ESI†). The 651 ppm signal is broadened due to coupling with both inner and outer Cd ions. ¹² After 1 h at 100 °C, the 651 pm (inner Cd sites) signal undergoes a shift to 561 ppm, while the weaker signal at 567 ppm (inner Cd

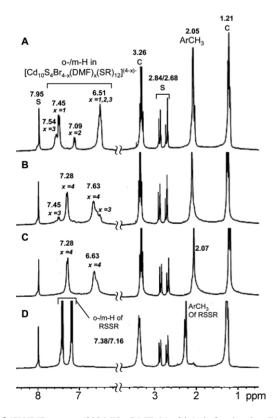


Fig. 3 ¹H NMR spectra (300 MHz, DMF-d₇) of **1** A) before heating, B) after 5 min at 100 °C, C) after 1 h at 100 °C, D) after addition of sulfur. S and C denote solvent and tetraethylammonium cation. The free thiol gives doublets at 7.16 and 7.04 ppm.

sites) can no longer be detected. This might be due to line broadening effects resulting from the coupling of ¹¹³Cd with the protons of DMSO.¹³ When a solution of elemental sulfur is added to the hot solution of 1, the absorption band at 380 nm in the UV/vis spectrum immediately shifts to its final position at 430 nm (spectrum C in Fig. 2), while the emission maximum of the solution in the PL spectrum moves to 496 nm (spectrum C in inset). These spectral features are consistent with the formation of CdS nanocrystals. The ¹H NMR spectrum in Fig. 3D reveals complete consumption of all cluster starting materials, and the formation of bis(4-toluyl)disulfide as the major byproduct. The identity of the disulfide was confirmed by comparison with that of an authentic sample of the material obtained by oxidation of the thiol with iodine in hexane. The data suggest that CdS nanocrystal growth is driven by oxidative removal of the thiolate ligands on the Cd₁₀S₄ cluster. This probably leads to unstable $Cd_{10}S_{4+x}(SR)_{12-2x}(DMF)_4$ fragments (x = 1-6) that subsequently condense into the CdS phase. These relatively soluble cluster fragments are likely to encapsulate the nanoparticle throughout the entire growth process and thus help to keep the nanoparticle in solution. This explains why no additional surfactants are needed in the synthesis. The efficiency of this cluster condensation process is not only illustrated by a 5.8 (1.2) nm single domain size, which corresponds to 930-3200 condensed Cd₁₀ clusters, but also by the low reaction temperature of 140 °C. This temperature is significantly below the temperature range routinely employed for the syntheses of CdS nanocrystals (250-300 °C) based on mononuclear (e.g. cadmium oxide and sulfur) starting materials, 14-16 indicating a different nanocrystal growth mechanism in our system. In conclusion we have demonstrated that crystalline CdS nanoparticles can be grown under mild conditions by reaction of cluster 1 with sulfur. This work confirms the value of molecular clusters as pre-assembled starting materials in the synthesis of inorganic nanocrystals.

Notes and references

‡ The oxidative aggregation of the Cd₁₀S₄ cluster with iodine has been previously investigated as an alternative pathway to CdS nanoparticles, but the products of this reaction were characterized only spectroscopically.¹⁷

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